

# Effects of Shear Rate, Temperature, and Polymer Composition on the Shear Stress of Polyethersulfone/1-Methyl-2-pyrrolidone Cast Solutions

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**ABSTRACT:** Polyethersulfone (PES) is extensively used for the preparation of hollow fiber membrane contactors; these modules are utilized in microfiltration (MF), ultrafiltration (UF), and gas separation. Viscosity plays a crucial role in the fabrication of hollow fiber membranes and their separation performance. In this study the effect shear rate, temperature, and polymer composition on shear stress of dope solutions prepared from PES polymer, dissolved in 1-methyl-2-pyrrolidone (NMP) solvent, is experimentally investigated. An empirical equation is developed to describe these effects on the shear stress of the polymer solutions. Data were obtained for (0.050, 0.10, 0.15, 0.20, and 0.25) mass fractions of PES polymer in NMP solvent at temperatures for (298, 308, 318, 328, 338, and 353) K. The shear stress results calculated by the developed equation describe well the set of experimental data.

## INTRODUCTION

Viscosity is an essential property which plays a significant function in fluid transportation, mixing, heat transfer, or mass transfer operations. Polymer solution viscosities at different temperatures are also important in polymer extruding and fabrication of polymeric hollow fiber membranes used in dialysis, water treatment, and gas separation processes. Fluids are classified based on their performance into Newtonian and non-Newtonian. Newtonian fluids are defined as those exhibiting a direct proportionality relationship between shear stress and shear rate, whereas for non-Newtonian fluids, the relationship between shear stress and shear rate is not linear. Pseudoplastic fluids are called shear thinning fluids, because their apparent viscosity decreases with shear rate; hence, increased shearing breaks down the internal structure within the fluid rapidly.<sup>1</sup>

The viscosity of polyisoimide solutions was studied by Levinson et al.;<sup>1</sup> two models were developed, to describe the effect of temperatures and shear rates on the viscosity. The data were obtained for various polyisoimide solutions prepared in different ratios of oxolane/diglyme in 1-methyl-2-pyrrolidone (NMP), and the viscosity of the polymer solutions was tested at various temperatures. These models account for temperature dependencies in the power law shear-thinning exponent. Experimental data were found to fit the proposed model reasonably well. The effects of polymer concentration, temperature, and surfactant on the rheological properties of poly(*n*-isopropylacrylamide), poly(NIP-AM), were studied by Tam et al.;<sup>2</sup> the investigators found that there do exist unusual rheological behavior for the effect of viscosity as a function of temperature which was elucidated to the phase behavior of the polymer. Benchabane and Bekkour<sup>3</sup> measured the critical concentrations of sodium carboxymethyl cellulose (CMC) with a rotational rheometer, and they observed that above a critical shear rate all CMC solutions exhibited shear thinning behavior and all of the flow curves were described by the cross model. The characterization

**Table 1. Statistics Used to Compare Shear Stress Calculated from eq 1 and Experimental Data**

|       |      |
|-------|------|
| AAD % | 1.2  |
| BIAS  | -5.1 |
| SDV   | 11   |
| rms   | 0.12 |

**Table 2. Estimated Parameters of eq 1**

| parameter         | estimated value | asymptotic standard error (ASE) | Wald 95 % confidence interval |       |
|-------------------|-----------------|---------------------------------|-------------------------------|-------|
|                   |                 |                                 | lower                         | upper |
| A/Pa              | 2.2             | 0.012                           | 2.2                           | 2.3   |
| B/s <sup>-1</sup> | 18000           | 120                             | 18000                         | 19000 |
| C/K               | 1500            | 5.2                             | 1500                          | 1600  |
| D                 | 0.28            | 0.021                           | 0.28                          | 0.29  |
| m                 | 1.3             | 0.013                           | 1.3                           | 1.3   |

of CMC solutions was completed with a time-dependent viscosity that showed that CMC solutions exhibit strong thixotropic behavior, mainly at high CMC concentrations. The influences of shear rate on relative viscosity of different polymer ratios of carboxymethyl cellulose and  $\kappa$ -carrageenan in aqueous solution were examined by Gómez-Díaz et al.,<sup>4</sup> and the effect of temperatures on rheological behavior of the solution was investigated.

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Table 3. Shear Stress,  $\tau$ , versus Shear Rate,  $\dot{\gamma}$ , at Different Temperatures and PES Mass Fractions

| 100W | $\dot{\gamma}/s^{-1}$ | $\tau/Pa$ |       |       |       |       |       |
|------|-----------------------|-----------|-------|-------|-------|-------|-------|
|      |                       | 298 K     | 333 K | 318 K | 328 K | 338 K | 353 K |
| 25   | 0.3                   | 105       | 85    | 64    | 49    | 42    | 30    |
|      | 0.7                   | 182       | 162   | 125   | 97.6  | 80.6  | 60    |
|      | 1                     | 250       | 231   | 178   | 141   | 116   | 87    |
|      | 1.4                   | 330       | 293   | 228   | 182   | 152   | 114   |
|      | 1.7                   | 390       | 353   | 276   | 222   | 185   | 139   |
|      | 2.1                   | 450       | 409   | 323   | 261   | 218   | 165   |
|      | 2.4                   | 500       | 464   | 368   | 298   | 249   | 189   |
|      | 2.8                   | 560       | 516   | 410   | 333   | 279   | 212   |
|      | 3.1                   | 620       | 569   | 450   | 367   | 307   | 234   |
|      | 3.5                   | 666       | 613   | 489   | 400   | 336   | 256   |
|      | 3.8                   | 717       | 658   | 527   | 433   | 364   | 278   |
|      | 4.1                   | 768       | 702   | 564   | 464   | 391   | 300   |
|      | 4.5                   | 817       | 744   | 600   | 495   | 416   | 321   |
|      | 4.8                   | 867       | 785   | 634   | 525   | 442   | 341   |
|      | 5.2                   | 916       | 824   | 668   | 554   | 467   | 361   |
|      | 5.5                   | 963       | 862   | 701   | 583   | 492   | 382   |
|      | 5.9                   | 1010      | 900   | 733   | 611   | 516   | 401   |
|      | 6.2                   | 1060      | 936   | 764   | 638   | 539   | 420   |
|      | 6.6                   | 1090      | 971   | 794   | 665   | 562   | 440   |
|      | 6.9                   | 1140      | 1010  | 825   | 690   | 585   | 457   |
| 7.2  | 1180                  | 1040      | 854   | 717   | 607   | 476   |       |
| 7.6  | 1220                  | 1070      | 883   | 742   | 629   | 494   |       |
| 7.9  | 1270                  | 1100      | 911   | 768   | 650   | 513   |       |
| 8.3  | 1325                  | 1130      | 938   | 792   | 672   | 530   |       |
| 8.6  | 1368                  | 1160      | 966   | 816   | 692   | 547   |       |
| 9    | 1411                  | 1190      | 1006  | 839   | 713   | 564   |       |
| 9.3  | 1454                  | 1220      | 1035  | 863   | 730   | 581   |       |
| 9.7  | 1496                  | 1240      | 1066  | 887   | 752   | 597   |       |
| 10   | 1520                  | 1290      | 1096  | 909   | 772   | 613   |       |
| 20   | 2                     | 87        | 71    | 56    | 44    | 36    | 27    |
|      | 3                     | 164       | 135   | 108   | 87    | 72    | 54    |
|      | 5                     | 234       | 194   | 156   | 126   | 105   | 79    |
|      | 7                     | 297       | 248   | 201   | 164   | 137   | 104   |
|      | 9                     | 356       | 299   | 243   | 199   | 167   | 127   |
|      | 10                    | 412       | 347   | 284   | 234   | 197   | 150   |
|      | 12                    | 464       | 393   | 322   | 266   | 225   | 172   |
|      | 14                    | 514       | 436   | 359   | 298   | 251   | 194   |
|      | 16                    | 562       | 478   | 394   | 329   | 278   | 215   |
|      | 17                    | 608       | 519   | 429   | 358   | 303   | 235   |
|      | 19                    | 651       | 558   | 462   | 387   | 328   | 255   |
|      | 21                    | 694       | 596   | 494   | 415   | 351   | 274   |
|      | 22                    | 734       | 632   | 525   | 442   | 375   | 294   |
|      | 24                    | 773       | 667   | 556   | 469   | 398   | 312   |
|      | 26                    | 811       | 701   | 585   | 495   | 420   | 330   |
|      | 28                    | 848       | 734   | 614   | 520   | 441   | 348   |
|      | 29                    | 884       | 767   | 642   | 545   | 463   | 366   |
|      | 31                    | 918       | 798   | 669   | 570   | 483   | 383   |
| 33   | 952                   | 829       | 696   | 594   | 504   | 400   |       |
| 35   | 985                   | 859       | 722   | 617   | 524   | 417   |       |
| 36   | 1020                  | 888       | 748   | 640   | 544   | 433   |       |
| 38   | 1050                  | 917       | 773   | 662   | 563   | 449   |       |

Table 3. Continued

| 100W | $\dot{\gamma}/s^{-1}$ | $\tau/Pa$ |       |       |       |       |       |
|------|-----------------------|-----------|-------|-------|-------|-------|-------|
|      |                       | 298 K     | 333 K | 318 K | 328 K | 338 K | 353 K |
| 15   | 40                    | 1095      | 945   | 797   | 684   | 581   | 465   |
|      | 41                    | 1132      | 972   | 821   | 706   | 600   | 481   |
|      | 43                    | 1170      | 998   | 845   | 727   | 618   | 496   |
|      | 45                    | 1203      | 1020  | 868   | 748   | 636   | 511   |
|      | 47                    | 1242      | 1050  | 890   | 769   | 654   | 526   |
|      | 48                    | 1275      | 1070  | 913   | 789   | 671   | 541   |
|      | 50                    | 1313      | 1100  | 934   | 809   | 688   | 556   |
|      | 35                    | 238       | 200   | 170   | 145   | 127   | 104   |
|      | 69                    | 400       | 344   | 293   | 251   | 218   | 176   |
|      | 104                   | 560       | 474   | 404   | 345   | 300   | 240   |
| 10   | 138                   | 700       | 593   | 504   | 431   | 374   | 302   |
|      | 172                   | 830       | 704   | 602   | 514   | 446   | 366   |
|      | 207                   | 960       | 815   | 692   | 593   | 514   | 421   |
|      | 241                   | 1080      | 918   | 780   | 669   | 580   | 472   |
|      | 276                   | 1200      | 1022  | 867   | 744   | 644   | 527   |
|      | 310                   | 1300      | 1110  | 940   | 800   | 704   | 578   |
|      | 345                   | 1400      | 1210  | 1025  | 870   | 766   | 620   |
|      | 379                   | 1530      | 1300  | 1110  | 940   | 820   | 670   |
|      | 409                   | 1600      | 1380  | 1150  | 1000  | 870   | 700   |
|      | 35                    | 51        | 41    | 35    | 29    | 25    | 17    |
| 5    | 69                    | 95        | 78    | 66.7  | 55.6  | 48    | 34.2  |
|      | 104                   | 135       | 112   | 96.1  | 80.9  | 70    | 50.5  |
|      | 138                   | 171       | 142   | 123   | 104   | 90    | 66.3  |
|      | 172                   | 204       | 171   | 149   | 127   | 110   | 81.6  |
|      | 207                   | 235       | 199   | 173   | 148   | 128   | 95.8  |
|      | 241                   | 265       | 224   | 196   | 169   | 146   | 110   |
|      | 276                   | 293       | 248   | 218   | 188   | 163   | 123   |
|      | 310                   | 319       | 272   | 240   | 207   | 179   | 137   |
|      | 345                   | 344       | 294   | 260   | 225   | 195   | 150   |
|      | 379                   | 368       | 316   | 279   | 242   | 211   | 162   |
|      | 414                   | 392       | 336   | 298   | 259   | 226   | 175   |
|      | 448                   | 414       | 356   | 317   | 276   | 242   | 187   |
|      | 483                   | 445       | 376   | 334   | 292   | 256   | 198   |
|      | 517                   | 465       | 395   | 352   | 308   | 270   | 210   |
|      | 552                   | 485       | 413   | 369   | 323   | 285   | 221   |
|      | 586                   | 510       | 431   | 385   | 338   | 299   | 233   |
|      | 621                   | 530       | 449   | 401   | 353   | 312   | 244   |
|      | 655                   | 560       | 465   | 417   | 367   | 325   | 255   |
| 690  | 580                   | 490       | 432   | 381   | 339   | 266   |       |
| 724  | 610                   | 510       | 447   | 395   | 352   | 276   |       |
| 759  | 630                   | 530       | 462   | 408   | 365   | 287   |       |
| 793  | 650                   | 544       | 476   | 421   | 378   | 297   |       |
| 828  | 680                   | 560       | 491   | 435   | 391   | 308   |       |
| 862  | 700                   | 580       | 505   | 447   | 404   | 318   |       |
| 897  | 720                   | 600       | 518   | 460   | 416   | 328   |       |
| 931  | 740                   | 620       | 532   | 473   | 429   | 339   |       |
| 966  | 760                   | 640       | 545   | 485   | 442   | 349   |       |
| 1000 | 780                   | 660       | 558   | 497   | 454   | 359   |       |
| 35   | 9                     | 6         | 5     | 4     | 4     | 3     |       |
| 69   | 17                    | 14        | 11    | 9     | 8     | 6     |       |
| 104  | 25                    | 20        | 16    | 13    | 12    | 10    |       |
| 138  | 32                    | 26        | 22    | 17    | 15    | 12    |       |
| 173  | 39                    | 32        | 27    | 21    | 19    | 15    |       |

Table 3. Continued

| 100W | $\dot{\gamma}/s^{-1}$ | $\tau/Pa$ |       |       |       |       |       |
|------|-----------------------|-----------|-------|-------|-------|-------|-------|
|      |                       | 298 K     | 333 K | 318 K | 328 K | 338 K | 353 K |
|      | 207                   | 47        | 38    | 32    | 25    | 23    | 18    |
|      | 241                   | 54        | 44    | 37    | 29    | 26    | 21    |
|      | 276                   | 62        | 50    | 42    | 33    | 30    | 24    |
|      | 310                   | 69        | 56    | 47    | 37    | 34    | 27    |
|      | 345                   | 76        | 62    | 52    | 41    | 37    | 30    |
|      | 379                   | 83        | 68    | 57    | 45    | 41    | 33    |
|      | 414                   | 90        | 73    | 61    | 49    | 45    | 36    |
|      | 448                   | 96        | 79    | 66    | 53    | 48    | 39    |
|      | 483                   | 103       | 85    | 71    | 57    | 52    | 42    |
|      | 517                   | 110       | 90    | 76    | 61    | 55    | 45    |
|      | 552                   | 116       | 96    | 81    | 65    | 59    | 48    |
|      | 586                   | 123       | 101   | 85    | 69    | 62    | 50    |
|      | 621                   | 130       | 107   | 90    | 73    | 66    | 53    |
|      | 655                   | 136       | 113   | 95    | 76    | 69    | 56    |
|      | 690                   | 142       | 118   | 100   | 80    | 73    | 59    |
|      | 724                   | 149       | 124   | 105   | 84    | 76    | 62    |
|      | 759                   | 155       | 129   | 109   | 88    | 80    | 65    |
|      | 793                   | 161       | 134   | 114   | 92    | 83    | 68    |
|      | 828                   | 167       | 140   | 118   | 95    | 87    | 71    |
|      | 862                   | 173       | 145   | 123   | 99    | 90    | 73    |
|      | 897                   | 180       | 150   | 128   | 103   | 94    | 76    |
|      | 931                   | 185       | 155   | 132   | 107   | 97    | 79    |
|      | 966                   | 192       | 161   | 137   | 111   | 101   | 82    |
|      | 1000                  | 197       | 166   | 141   | 114   | 104   | 85    |

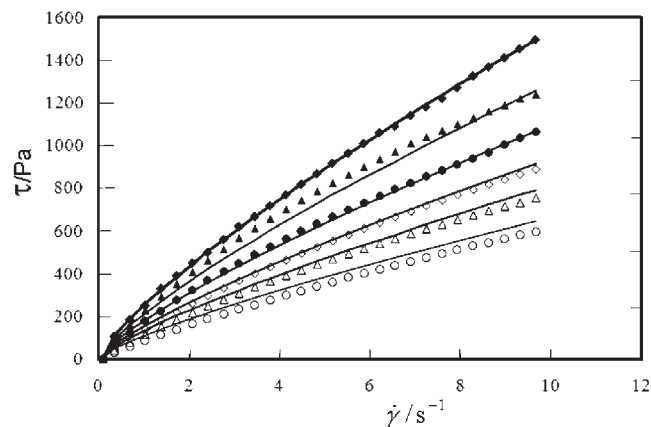


Figure 1. Shear stress versus shear rate at various temperatures, 0.25 mass fractions PES. Solid lines are calculated values.  $\blacklozenge$ , 298 K;  $\blacktriangle$ , 308 K;  $\bullet$ , 318 K;  $\diamond$ , 328 K;  $\triangle$ , 338 K;  $\circ$ , 353 K.

Models based on the viscosity of individual polymer solutions were engaged to check up the experimental data.

In general, viscosity is an essential factor in the fabrication of polymeric hollow fiber membranes. Among these polymers, polyethersulfone (PES) is widely used for the preparation of microfiltration (MF), ultrafiltration (UF), and gas separation hollow fiber and flat sheet membranes. PES hollow fiber membranes are promising technology for water treatment and removal of pollutant gases such as  $CO_2$  and  $H_2S$  from natural gas and flue gas. Membrane ultra filtration is

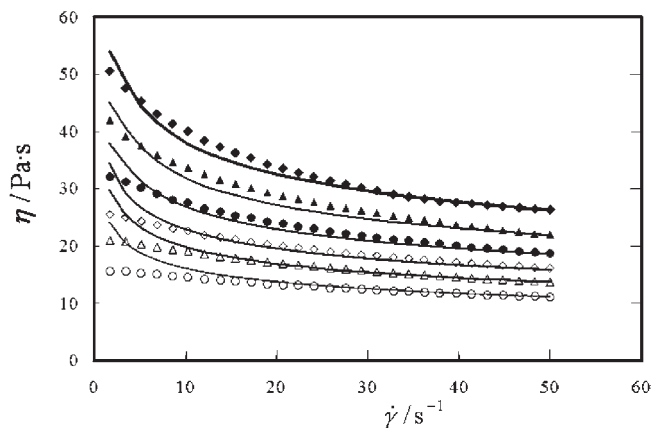


Figure 2. Viscosity versus shear rate at various temperatures, 0.20 mass fractions PES. Solid lines are calculated values.  $\blacklozenge$ , 298 K;  $\blacktriangle$ , 308 K;  $\bullet$ , 318 K;  $\diamond$ , 328 K;  $\triangle$ , 338 K;  $\circ$ , 353 K.

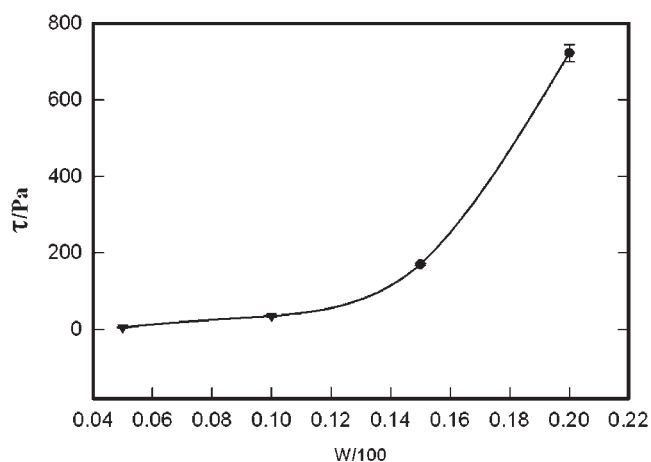
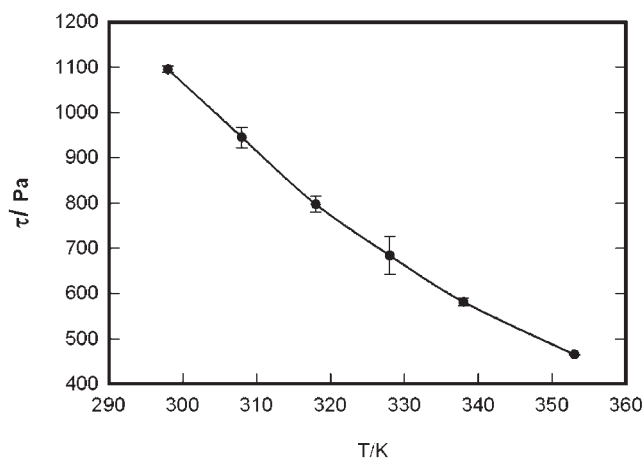


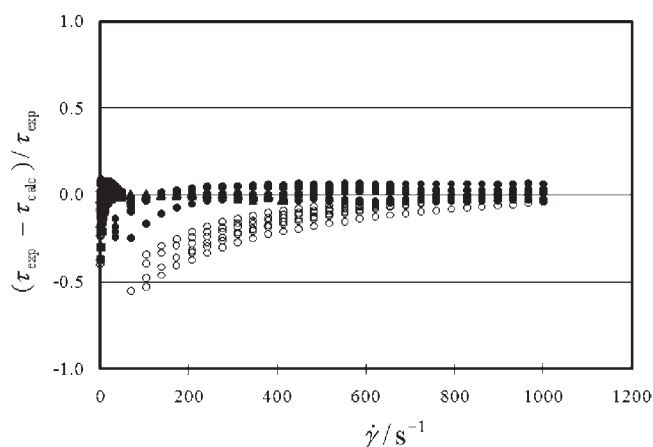
Figure 3. Shear stress versus PES mass fractions at 318 K and shear stress  $35 s^{-1}$ . The solid line is the calculated values.

well-known as a capable technique in water purification process. Membrane filtrations are easy to operate in comparison with conventional water treatment methods, and the energy required for operation and maintenance is small. It is a talented technology for purification and production of drinking water. Membrane filtration methods are capable of disinfecting water and removing its turbidity at moderately low pressure. Another advantage of membrane filtrations are the capability of removing a broad range of substances, and the production of stable quality water.<sup>5-11</sup>

As viscosity is an important factor in the preparation of PES hollow fibers membranes, the aim of the present work is to experimentally study the effect of shear rate, temperature, and polymer compositions on shear stress and to develop a model equation capable to describe the effect of shear rate, temperature, and PES mass fraction on shear stress. The developed model equation can also be used to predict shear stress for polymer solutions at various PES mass fractions and temperatures that have not been experimentally studied and are within the correlation acceptable range. Various solutions of PES/NMP mass fractions (0.050, 0.10, 0.15, 0.20, and 0.25) are prepared and tested at various temperatures (298 to 353) K and variable shear rates.



**Figure 4.** Shear stress versus temperature at fixed PES mass fraction, 0.20, and fixed shear rate,  $35 \text{ s}^{-1}$ .



**Figure 5.** Fractional deviation between experimental ( $\tau_{\text{exp}}$ ) and calculated ( $\tau_{\text{calc}}$ ) shear stress versus shear rate for various PES mass fractions;  $\diamond$ , 0.25;  $\blacksquare$ , 0.20;  $\blacktriangle$ , 0.15;  $\bullet$ , 0.10;  $\circ$ , 0.05.

## EXPERIMENTAL SECTION

PES (Ultrason E6020 P), with a weight-average molecular weight of 65 000, was purchased from BASF Company. The solvent, 1-methyl-2-pyrrolidone (purity 0.99 mass fraction), was obtained from Wako Pure Chemical Industries, Ltd. (Osaka, Japan). Ultrason pellets can absorb moisture very rapidly and must be dried before processing. Vacuum or dry air oven operating at (403 to 423) K is recommended. Circulating air ovens are unsuitable. Drying time is dependent on moisture level, but the materials must be dried at least four hours. Five different concentrations of PES/NMP were prepared: (0.05, 0.10, 0.15, 0.20, and 0.25) mass fraction PES. Each solution was mixed for 24 h using magnetic stirrer to ensure the solutions are homogeneous and perfectly mixed. The rotational viscometer, Rheolab QC, from Anton Paar, Austria [maximum torque of  $50 \text{ mN}\cdot\text{m}$ , torque resolution of  $0.01 \text{ mN}\cdot\text{m}$ , speed ( $0.6$  to  $90000$ )  $\text{s}^{-1}$ , shear stress range of ( $0.25$  to  $75$ )  $\text{mN}\cdot\text{m}$ , shear rate range of ( $10^{-2}$  to  $4000$ )  $\text{s}^{-1}$ , viscosity measuring range of ( $1$  to  $10^9$ )  $\text{mPa}\cdot\text{s}$ ] is used for the shear stress measurements. In this instrument, the viscous torque on a spinning object is monitored by measuring the rotation frequency of the object. This is converted directly into a

viscosity by the software of the instrument and reported on the screen. The device was used to measure the shear stress of the polymer solutions versus shear rate for temperatures in the range (298 to 353) K; the same range of temperatures is used in the preparation of polymeric hollow fiber membrane utilizing nonsolvent-induced face separation techniques.

**Correlation Development.** Most of the equations which describe the relationship between the shear stress and shear rate are empirical correlations. The two-parameter power-law models ( $\tau = K\dot{\gamma}^n$ ) are the most commonly used as an empirical equation for pseudoplastic fluids. The power law model is a useful rheological model that describes the relationship between viscosity or shear stress and shear rate in a specific range of shear rates where shear thinning occurs in a non-Newtonian fluid. The power law is accurate for the prediction of viscosity at various shear rates for molten polymers.<sup>12</sup> The applicability of the power law gets better with the raise of solution concentrations. If  $n$  is equal to 1, the flow is Newtonian, and the viscosity does not change with shear rate. The flow is pseudoplastic or shear thinning if  $n$  is less than 1. Most polymer melts and solutions are pseudoplastic. The flow is dilatant or shear-thickening if  $n$  is greater than 1. The Andrade equation describes the temperature reliance of liquid viscosities.<sup>13</sup> In the present work, a more comprehensive model adopted from the parabolic shape is used in fitting the experimental data.<sup>14,15</sup>

$$\tau = A[(1 + B^{-1}\dot{\gamma})^m - 1]^{m-1} e^{CT^{-1}} e^{D(100W)} \quad (1)$$

where  $\tau/\text{Pa}$  is the shear stress,  $\dot{\gamma}/\text{s}^{-1}$  is the shear rate, and  $A/\text{Pa}$ ,  $B/\text{s}^{-1}$ ,  $C/\text{K}$ ,  $D$ , and  $m$  are empirical parameters.  $T/\text{K}$  is the temperature, and  $W$  is the mass fraction of PES in NMP.

In this correlation  $m$  is just a fitting parameter and is different than  $n$  in the power law model. The empirical parameters are obtained by fitting the experimental data with the proposed model. The statistics used to compare shear stress calculated using eq 1 to experimental shear stress can be found elsewhere.<sup>16</sup>

High values of the average absolute deviation (AAD) and BIAS indicate systematic differences between experimental data and calculated results. Values of the standard deviation (SDV) give an indication of the systematic or random dispersion of the data set about the BIAS value. The rms provides another indication of the systematic or random dispersion of the data from the developed equation. Data sets are accurately represented by the developed correlation when all four statistical parameters are near zero.

The fitted empirical constants were calculated using Easy-Fit software package.<sup>17,18</sup> The statistics was used to compare shear stress calculated from eq 1 and experimental data shown in Table 3. The estimated parameters were within the 95 % confident interval lower and upper limits. The obtained  $R^2$  values were relatively high (above 0.99). The AAD, SDV, BIAS, and root-mean-square (rms) were calculated and are shown in Table 1. As close the  $|\text{Bias}|$  is to the rms, the bigger are the systematic deviations. Bias = 0 means that there are not systematic deviations. As Bias is more or less the half of the rms, the systematic deviations are of the same magnitude than the random deviations. Estimated values of ADD, SDV, BIAS, and rms are not very close to zero which indicates that there is a systematic deviation. Figure 5 shows that the systematic deviations are quite important mainly for low PES mass fractions of 0.05. The parameters of eq 1 are shown in Table 2. The experimental data at various shear rates, temperature, and PES compositions are shown in Table 3.



## RESULTS AND DISCUSSION

Figure 1 shows the effect of shear stress versus shear rate for 0.25 PES mass fractions in NMP solvent at five different temperatures: (298, 308, 318, 328, 338, and 353) K. The figure demonstrates that, as shear rate increases, shear stress increases accordingly at any of those temperatures. At a specific value of shear rate, the shear stress was found to increase exponentially as polymer mass fraction increases, because, the polymer solution becomes thicker when the amount of polymer added to a fixed volume of solvent, increases. Calculated values were close to the experimental data at high values of shear rates; by contrast, at low shear rates there is significant deviation. The effect of solution viscosity of 0.2 mass fractions PES at variable shear rates and five different temperatures is depicted in Figure 2. The figure revealed how the viscosity is affected with changes in shear rate, temperature, and polymer compositions; at high shear rates the change in viscosity is very insignificant. At fixed values of shear rate, the viscosity decreases as temperature increases.

The effect of polymer mass fraction on shear stress at a fixed temperature (308 K) and fixed shear rate ( $35 \text{ s}^{-1}$ ) is revealed in Figure 3. The figure exemplifies the exponential relationship between shear stress and polymer mass fractions. This backs up the exponential relationship between shear stress and polymer mass fraction used in eq 1. The shear stress increased sharply for solutions with mass fractions higher than 0.15. This confirmed the suitable range of PES mass fraction in dope solutions used in the fabrication of polymeric hollow fiber membranes (the PES/NMP suitable range is between 0.15 and 0.25 mass fractions of PES). The effect of temperature on shear stress is depicted in Figure 4; the figure demonstrates that shear stress decreases as temperature increases. This agreed with the inverse relationship between shear stress and temperature used in eq 1. At low PES mass fractions, polymer solution shows evidence of Newtonian behavior fluid. The fractional deviation between experimental and calculated shear stress versus shear rate is shown in Figure 5. The plot shows that the deviation between calculated and experimental values is considerable at the low shear rates (less than  $200 \text{ s}^{-1}$ ) and for a PES mass fraction of 0.05. By contrast, the discrepancy decreases with increasing shear rates and higher values of PES mass fractions.

## CONCLUSIONS

A generalized model equation was developed to depict shear stress as a function of shear rate, temperature, and polymer mass fractions for PES/NMP polymer solutions. The polymeric solution is frequently used in the fabrication of hollow fiber membrane using nonsolvent-induced phase separation (NIPS) techniques. The generated correlation successfully describes shear stress to the shear rate for shear rate range ( $0$  to  $1000 \text{ s}^{-1}$ ), and PES mass fractions range from (0.05 to 0.25). The PES/NMP polymeric solution behaves as pseudoplastic fluids, the shear stress increases as shear rate increases, and polymer mass fractions in the dope solution increases. The developed equation is reliable to predict shear stress at various shear rates, temperature, and polymer mass fraction for values other than existing data.

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